

Keywords: hybrid flax; supersap epoxy; mechanical properties; environmental resistance

composites[32]. However in the present investigation, the treatments had little effect on (increasing T_g through better interfacial adhesion. This primarily is attributed to the decrease in fiber strength, leading to fewer constraints on segmental flexibility. For the same reason, the highest and lowest loss modulus was found in untreated and BTCA treated composites with the maximum and minimum viscous dissipation, respectively.

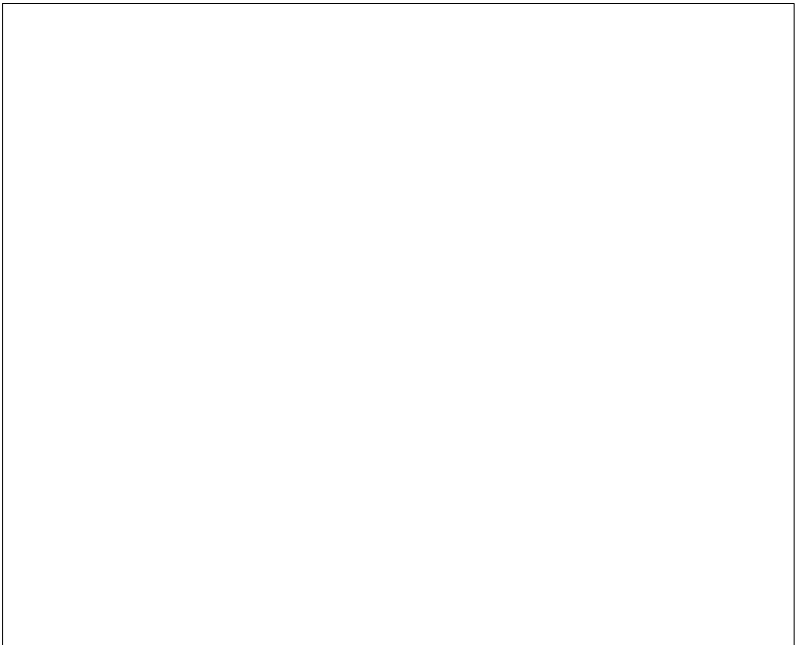


Figure 3. Effect of treatments on loss modulus of resin and FE composites.

Table 4. Thermalmechanical properties of untreated and treated composites

Testing material	T_g (Loss) (°C)	T_g	7 D Q /	Max E $\dot{\epsilon}$ (MPa)	Max Tan /	B
Pure bioepoxy	72.74		84.83	241	0.834	0
Untreated	77.32		82.43	430	0.144	1.69
NaOH treated	75.28		81.29	400	0.176	1.61
BTCA treated	75.49		82.00	337	0.188	1.58
APS treated	76.79		81.87	356	0.155	1.66
LD treated	75.55		82.30	420	0.161	1.64

The ratio of storage (elastic) and loss (viscous) modulus called tan δ (Figure 4). In comparison to the untreated composites, an increase in magnitude of tan δ peak was observed in treated flax/supersap composites. Thus, it will tend to dissipate less energy with lower magnitude of damping peak in FRP. In the present study, the untreated composites showed a rebound case subjected to the falling weight impact, while the treated composites broke longitudinally. The untreated FE composites had better damping property, thus dissipating more energy by internal friction.



Figure 4. Representation of a flax/supersap composite with treated flax fibers.

The bonding properties between fibres could also be indicated quantitatively. According to the following equation, the bonding properties for various treated flax composites could be assessed by [24]:

$$B = \frac{F_a - f_a}{L_d} \quad (1)$$

where B is a parameter used to indicate the interfacial bonding strength for composites and f_a for pure matrix. V_d is the fibre volume fraction which could be calculated from composite volume V_c , composite weight (M_c), fiber weight ratio (W_f) and matrix weight ratio (W_m):

$$V_d = \frac{F_a H : s F_d}{Q_a} \quad (2)$$

The lower value of B parameter indicates lower interfacial strength. From the results summarized in Table 4, B value decreased after all the treatments, possibly owing to the loss of matrix of fibre (strength).

3.2. Effect of Treatment on Quasi-Static Tension Properties

The tensile properties of flax/supersap composites were compared through different treatments (non-treated, NaOH, BTCA, APS and Laccase). As shown in Figure 5, the untreated flax/supersap composites exhibited tensile strength and tensile modulus of up to 185 MPa and 14 GPa, respectively. The NaOH treated composites displayed the tensile strength of about 178 MPa, less than 10 MPa reductions in tensile strength. The average tensile strength was observed to be at 175 MPa and 165 MPa for APS and Doga treated composites, respectively. Over 70% reduction of tensile strength was found for the BTCA treated flax/supersap laminates (51 MPa) compared to untreated composites (185 MPa). Clearly, these chemical treatments have no significant effect on improvement in ultimate tensile strength of the composites although they may have positive

Figure 7. Flax/supersap bio-epoxy composites samples with and without treatments after aging condition: a) UV radiation for 552 h; b) Xenon light for 500 h.

Table 5. Tension properties of hybrid flax/supersap bio-epoxy composites with different treatments in weathering conditions

Samples	Tensile Strength (MPa)			Tensile Modulus (GPa)		
	Normal	UV	Xenon	Normal	UV	Xenon
Untreated	185.4± 8.5	185.4± 7.8	172.2± 8.1	13.9± 0.4	14.0± 0.3	8.8± 0.4
NaOH	178.5± 6.4	146.9± 6.4	169.8± 6.5	11.9± 0.5	10.5± 0.4	11.8± 0.5
BTCA	51.7± 6.1	40.2± 5.8	45.9± 6.2	6.4± 0.6	5.4± 0.2	6.0± 0.2
APS	175.4± 6.2	157.3± 8.2	175.1± 6.2	11.3± 0.4	9.8± 0.5	11.5± 0.6
LD	164.5± 5.7	157.1± 7.0	160.3± 5.6	13.7± 0.4	13.7± 0.2	11.9± 0.3

4. Conclusions

The untreated new flax/epoxy composites (tensile strength of around 185 MPa) show a good potential to be used in automobile applications. Pure NaOH treatment is the most promising one among treatments performed in this study. NaOH composites maintained the tensile strength of around 180 MPa; however, BTCA and other treatments decreased the tensile strength. The improvement in fibre/matrix wettability may not be able to compensate the reduction in tensile strength (due to the removal of interfibrillar matrix by general treatments). These treatments had little effect on glass transition temperature (approximately 75 °C).

This work also suggests that the applied treatments for flax/supersap bio-epoxy composites (57 wt% fibre ratio) can significantly improve Xenon aging resistance, but have little effect on UV aging. For UV and Xenon aging, the composites with modified flax did not show any visual change, which was a bit better than composites without flax modification. With respect to the tensile properties, the untreated composites had the best performance after UV exposure, while for Xenon exposure; the four treated composites (NaOH, BTCA, APS and LD) showed much better resistance.

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Author Contributions

The Jinchun Zhu designed the study, performed the experiments and analysis, and wrote the first draft of the manuscript. Other authors aided the analyses of the study.

Conflicts of Interest

The authors declare no conflict of interests. The funding sponsors had roles in the design of the study and in the decision to publish results.

References

1. Zhu, J.; Abhyankar, H.; Nassiopoulous, E.; Njuguna, J. Tabased flax fibre reinforced composites for structural applications in vehicles IOP Conf. Ser. Mater. Sci. Eng. 2012, 40, 012030.
2. Avril, C.; Bailly, P.A.; Njuguna, J.; Nassiopoulous, E.; Larminat, D.A. Development of Flax-Reinforced BioComposites for High Load Bearing Automotive Parts. In Proceedings of the European Conference on Composite Materials (ECCM), Venice, Italy, 24–28 June 2012.
3. Fan, J.; Nassiopoulous, E.; Brighton, J.; de Larminat, A.; Njuguna, J. New Structural Biocomposites for Car Applications. In Proceedings of the Society of Plastics Engineers EUROTEC 2011 Conference Proceedings, Barcelona, Spain, 3–5 October 2011.
4. Adekunle, K.; Cho, S.W.; Ketzscher, R.; Skrifvars, M. Mechanical properties of natural fiber hybrid composites based on renewable thermoset resins derived from soybean oil, for use in technical applications. Appl. Polym. Sci. 2012, 124, 4530–4541.
5. Adekunle, K.; Cho, S.W.; Patzelt, C.; Blomfeldt, T.; Skrifvars, M. Impact and flexural properties of flax fabrics and Lyocell fibre reinforced bio-based thermoset. Reinf. Plast. Compos. 2011, 30, 685–697.
6. Lincoln, J.D.; Shapiro, A.A.; Earthman, J.C.; Saphores, M.; Ogunseitan, O.A. Design and evaluation of bioepoxy/flax composites for printed circuit boards. IEEE Trans. Electron. Packag. Manuf. 2008, 31, 211–220.
7. Berger, C.; Bledzki, A.K.; Heim, H.-P.; Bötcher, A. Fiber Reinforced Epoxy Composites Made from Renewable Resources. In Proceedings of the International SAMPE Technical Conference, Fort Worth, TX, USA, 17–21 October 2011.
8. Felling, F.; Pappadà, S.; Gennaro, R.; Passaro, R. Resin transfer moulding of composite panels with bio-based resins. SAMPE J. 2013, 49, 20–24.
9. Huang, X.; Netravali, A. Characterization of flax fiber reinforced soy protein resin based green composites modified with nanoclay particles. Compos. Sci. Technol. 2007, 67, 2005–2014.

10. Oksman, K. High quality flax fibre composites manufactured by the resin transfer moulding process. *J. Reinf. Plast. Compos.* 2001, 20, 621–627.
11. Zhu, J.; Njuguna, J.; Abhyankar, H.; Zhu, H.; Perreux, D.; Thiebaud, F.; Chapelle, D.; Apiz Sauget, A.; de Larminat, A.; Nicollin, A. Effect of fibre configurations on mechanical properties of flax/tannin composites. *Ind. Crops Prod.* 2013, 50, 68–76.
12. Zhu, J.; Abhyankar, H.; Njuguna, J. Effect of Fibre Treatment on Water Absorption and Tensile Properties of Flax/Tannin Composites. In Proceedings of the ICMR 2013 Cranfield, UK, 19–20 September 2013.
13. Summerscales, J.; Dissanayake, N.P.J.; Virk, A.S.; Hall, W. A review of bast fibres and their composites. Part 1: Fibres as reinforcement. *Compos. Part A Appl. Sci. Manuf.* 2010, 41, 1329–1335.
14. Cantero, G.; Arbelaiz, A.; Llanillo, R.; Mondragon, I. Effects of fibre treatment on wettability and mechanical behaviour of flax/polypropylene composites. *Compos. Sci. Technol.* 2003, 63, 1247–1254.
15. Kaith, B.S.; Kalia, S. Grafting of flax fiber (*Linum usitatissimum*) with vinyl monomers for enhancement of properties of flax phenolic composites. *Polym. J.* 2007, 39, 1319–1327.
16. Hughes, M.; Carpenter, J.; Hill, C. Deformation and fracture behaviour of flax reinforced thermosetting polymer matrix composites. *Mater. Sci.* 2007, 42, 2499–2511.
17. Jähn, A.; Schröder, M.W.; Fühling, M.; Schenzel, K.; Diepenbrock, W. Characterization of alkali treated flax fibres by means of FT Raman spectroscopy and environmental scanning electron microscopy. *Spectrochim. Acta Part A Mol. Biomol. Spectrosc.* 2002, 58, 2271–2279.
18. Zhu, J.; Zhu, H.; Njuguna, J.; Abhyankar, H. Recent development of flax fibres and their reinforced composites based on different polymeric matrix. *Materials* 2013, 6, 5171–5198.
19. Yan, L.; Chouw, N.; Yuan, X. Improving the mechanical properties of natural fibre fabric reinforced epoxy composites by alkali treatment. *J. Reinf. Plast. Compos.* 2012, 31, 425–437.
20. Van de Weyenberg, I.; Chi Truong, T.; Vande, B.; Verpoest, I. Improving the properties of UD flax fibre reinforced composites by applying an alkaline fibre treatment. *Compos. Part A Appl. Sci. Manuf.* 2006, 37, 1368–1376.
21. Alix, S.; Lebrun, L.; Morvan, C.; Marais, S. Study of water behaviour of naturally treated flax fibres-based composites: A way to approach the hydric interfacial. *Compos. Sci. Technol.* 2011, 71, 893–899.
22. Bledzki, A.K.; Fink, H.-P.; Specht, K. Unidirectional hemp and flax-EP and PP composites: Influence of defined fiber treatments. *Appl. Polym. Sci.* 2004, 93, 2150–2156.
23. John, M.J.; Anandjiwala, R.D. Recent developments in chemical modification and characterization of natural fiber reinforced composites. *Polym. Compos.* 2008, 29, 187–207.
24. John, M.J.; Anandjiwala, R.D. Chemical modification of flax reinforced polypropylene composites. *Compos. Part A Appl. Sci. Manuf.* 2009, 40, 442–448.
25. Assarar, M.; Scida, D.; El Mahi, A.; Poilâne, C.; Ayad, R. Influence of water ageing on mechanical properties and damage events of reinforced composite materials: Fibres and glass fibres. *Mater. Des.* 2011, 32, 788–795.
26. Stamboulis, A.; Baillie, C.A.; Garkhail, S.K.; Van Melick, H.G.H.; Peijs, T. Environmental durability of flax fibres and their composites based on polypropylene matrix. *Appl. Compos. Mater.* 2000, 7, 273–294.

